



## Sputtering Synthesis and Thermal Annealing Effect on Gold Nanoparticles in Al<sub>2</sub>O<sub>3</sub> Matrix

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### ABSTRACT

Gold-Alumina nano composite thin films were prepared by RF-magnetron sputtering technique on glass substrate at room temperature. Subsequent thermal treatment was used to promote gold nano clusters formation. After annealing, an increase of size accompanied by an increase of the lattice constant parameter is observed. A value of the surface stress coefficient of gold is  $f = 3.11$  N/m has been obtained and a value of lattice thermal expansion coefficient of gold  $\alpha_{gold}(T) = 2.6 \times 10^{-5} / ^\circ\text{C}$  is achieved. Broad band absorption with maximum absorbance at around 500 nm characteristic of gold nanoclusters was observed after heat treatment. Experimental optical absorption spectra were modeled by the modified Mie theory. A red shift of the plasmon peak is obtained with increasing the annealing temperature.

### 1. Introduction

Materials tailored at the nanoscale, e.g. in the form of nanoparticles, have unique physical and chemical properties compared to their bulk form [1]. Among them, noble metals are the most studied plasmonic materials when they are embedded in a dielectric matrix they manifest surface plasmon resonance (SPR) in the visible range [2-4]. SPR is the unique property of noble metal nanoparticles due to its resonant collective oscillation of electrons in the presence of electromagnetic field. The SPR peak position wavelength can be tailored by varying the size, shape, inter particles separation, dielectric constant of metal nanoparticles as well as the embedding matrix which has been investigated by several researchers [5-9]. The ability to integrate metal nanoparticles into biological systems with general lack of toxicity has had greatest impact in biology and medicine. Among noble metal particles, gold nanoparticles have attracted intensive interest because they are easily prepared, have low toxicity and can be readily attached to molecules of biological interest [10]. An influence of the temperature on the SPR of metal nanoparticles is crucial for pure and applied science [11, 12]. The temperature dependence of the SPR resonance is important because of the recent applications of noble metal nanoparticles in thermally assisted magnetic recording [13] thermal cancer treatment [14-17] catalysis and nanostructure growth [18] and computer chips [19]. Tunability of SPR of noble metal nanoparticles have been achieved by incorporating them into oxide based matrix. Uniformly distributed in various oxide based matrices like SiO<sub>2</sub>, ZnO, MgO, TiO<sub>2</sub>, etc. are promising candidate for optical applications. Of all oxide based matrices, alumina oxide (Al<sub>2</sub>O<sub>3</sub>) is one of the most attractive material particularly because of its thermal and chemical stability, excellent dielectric properties, good adhesion to many surfaces and similar band gap (8.8 eV) to SiO<sub>2</sub> [20]. It makes a promising candidate for replacing SiO<sub>2</sub> for gate dielectrics. For these reasons, it is the mostly used host material. Y. Hosoya et al. [21] prepared Au-particle-doped Al<sub>2</sub>O<sub>3</sub> films by a sol-gel method and achieved the large nonlinear optical susceptibility and short relaxation time of the film. Garcia-Serrano et al. [22] prepared Au/Al<sub>2</sub>O<sub>3</sub> composite thin film by radio-frequency co-sputtering technique and found that the apparent optical absorption band due to the SPR of Au particles appeared and the intensity of the SPR absorption band characteristic of Au nanoparticles increased with the increase of Au particle size. Serna et al. [23] prepared the Cu/Al<sub>2</sub>O<sub>3</sub> nanocomposite thin films by alternate pulsed laser deposition from pure metallic Cu and ceramic Al<sub>2</sub>O<sub>3</sub> targets. The films showed a nonlinear optical response that is modified by laser irradiation.

A large number of methods have been used to obtain AuNPs embedded in Al<sub>2</sub>O<sub>3</sub> films such ion implantation [24-27], sol-gel [21, 28], Laser evaporation and cluster deposition [29-32], and RF magnetron sputtering [22, 33-35]. The flexibility and easy fabrication of diverse composite films are the advantages of sputtering method.

In this work, Au/Al<sub>2</sub>O<sub>3</sub> nanocomposite films were synthesized on glass substrate at room temperature by RF-magnetron sputtering technique. The as-deposited and heat-treated films were characterized by X-ray diffraction and optical absorption spectroscopy. Evolution of structural and optical properties of gold nanoparticles in alumina matrix as a function of temperature has been studied.

### 2. Experimental Methods

Au/Al<sub>2</sub>O<sub>3</sub> composite thin films were prepared by RF magnetron sputtering technique using an Alcatel SCM 650 apparatus. The target is constituted by two materials: pure metal Au chips (99.99%) on top of alumina disc of purity 99.99%. The gold to alumina surface ratio is fixed at 1.3%. The chamber was evacuated to a pressure better than 10<sup>-6</sup> mbar, before the argon gas for the sputtering was introduced. The deposition was accomplished on clean glass substrates at room temperature at a fixed argon pressure of 5x10<sup>-3</sup> mbar. The substrate to target distance, deposition time and applied power were fixed at 60 mm, 4h30min and 50w respectively. The prepared samples were annealed for 1 hour at different temperatures.

For analysis of the samples, the structural characterization was carried out using a Siemens D5000 diffractometer, employing CuK<sub>α</sub> radiation in the  $\theta/2\theta$  configuration. The optical transmission spectra were registered by a Shimadzu UV 3101 PC spectrophotometer in the wavelength range of 300 - 2000 nm.

### 3. Results and Discussion

#### 3.1 Structural Analysis

Fig. 1 shows the XRD patterns of the Au/Al<sub>2</sub>O<sub>3</sub> composite films before and after annealing. It can be seen that the spectra of all the samples had a shoulder around  $2\theta = 40^\circ$  and a broad peak at around  $64^\circ$ . The crystalline phases of the Au/Al<sub>2</sub>O<sub>3</sub> composite film were determined from a deconvolution procedure of the XRD spectrum with commercial software (program) available on our computer. Note that the purpose of the deconvolution is to fit the measured XRD spectrum in well-defined peaks to which a physical meaning can be attributed. For more details see the

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works [35-37]. After thermal treatment the peak intensities slightly increases evidencing some improvement in film crystallinity. Fig. 2 shows the experimental diffractogram and their curve fitting of the Au/Al<sub>2</sub>O<sub>3</sub> composite sample deposited at room temperature. Therefore, the XRD spectra of the composite film have to be deconvoluted with four peaks as illustrated in Fig. 2. The peak centered on 2θ = 26° corresponds to the amorphous alumina. For comparison, X-ray diffractogram of gold thin film with a cubic structure presented as a reference is also reported in Fig. 1. The peak positions are in agreement with the well-known data: JCPDS - 04 - 0784 characteristic of a gold crystal in a cubic phase, indicating that gold particles in different composite films should adopt an fcc-like structure. The peaks are considerably broadened; the extent of broadening is described by full width at half maximum intensity of the peak (FWHM). This broadening of the diffraction peak is related to the size nanoparticle D via the Scherrer's formula:  $D = 0.9\lambda / (FWHM)\cos(\theta_B)$ , where λ is the X-ray wavelength and θ<sub>B</sub> is the Bragg angle of the peak.

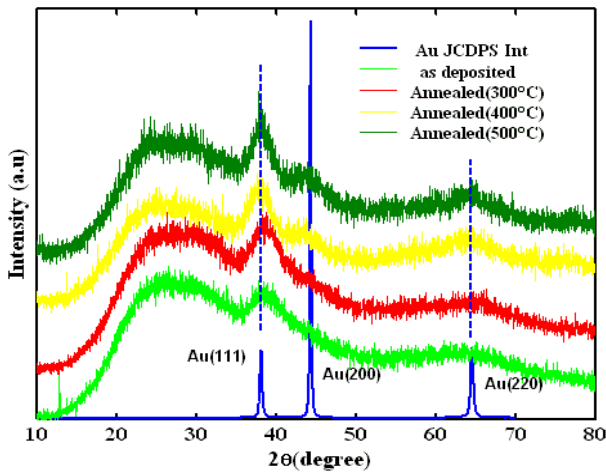


Fig. 1 X-ray diffractograms of Au/Al<sub>2</sub>O<sub>3</sub> nanocomposite thin films as-grown, heated at 300 °C, 400 °C, 500 °C and JCPDS of gold thin films

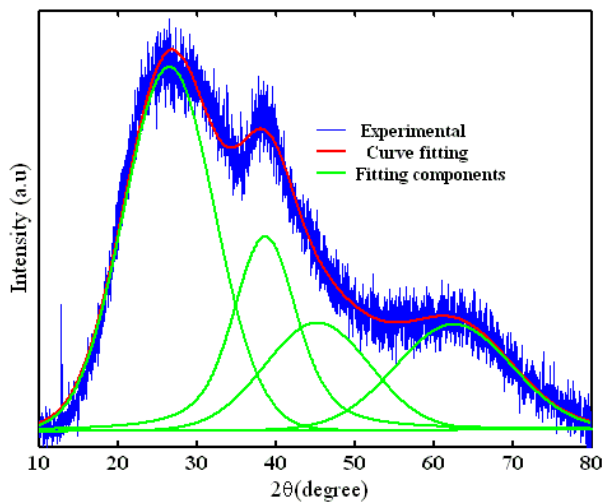


Fig. 2 Experimental diffractogram of the sample deposited at room temperature and their curve fitting where different pseudo-Voigt functions were taken into account

Table 1 Results of the curve fitting of the experimental diffractograms calculated from Au (111) reflections of the as-grown and annealed samples

Temperature (°C)	FWHM (degree)	2θ (degree)	Lattice parameter (Å)	Size(Å)
25	10.01	38.66	4.034	8.42
300	8.97	38.48	4.051	9.38
400	7.00	38.29	4.070	12.02
500	5.75	38.15	4.085	14.60

Table 1 summarizes the fitting parameters determined from the Au (111) orientation plane for all the samples. This samples indicates that Au NPs size shows an appreciable increase after heat treatment above 300 °C. Then, by varying the annealing temperature, we can control the size of Au nanoparticles. The size varied from 0.84 to 1.46 nm and the lattice constant parameter increased when heat temperature varies from 25 to 500 °C. So, lattice contraction depends implicitly on the particle size.

The concept of surface stress in solids, introduced by Gibbs [38] implies the occurrence of a hydrostatic compression induced by surface/interface stress leading to a change of the lattice parameter, Δa, as compared to the lattice parameter of the undistorted coarse-grained counterpart, a<sub>0</sub>, given by Mays et al [39]:

$$\Delta a = -\frac{4}{3D}kf a_0 \tag{1}$$

Where f is the surface/interface tension, D is the spherical grain diameter and k is the bulk modulus. With decreasing grain size, the effect of surface/interface stress will become significant and a lattice contraction is observed. Therefore, a measure of the lattice variation parameter Δa allows us to determine the surface stress f. The graphical representation of the lattice spacing versus the reciprocal diameter of gold nanoparticle is shown in Fig. 3. So, from the slope of the linear regression, the interface stress is evaluated according to the Eq. (1) with the value for the compressibility of bulk gold, K = 5.848 x 10<sup>-12</sup> Pa<sup>-1</sup> [40]. The value of f = 3.11 N/m has been obtained for gold clusters embedded in alumina matrix. A comparison to published experimental f coefficients shows that the determined surface stress value is lower than f = 3.83 N/m, determined by Solliard et al [41] and in good agreement with, f = 3.2 N/m, determined in the work [36].

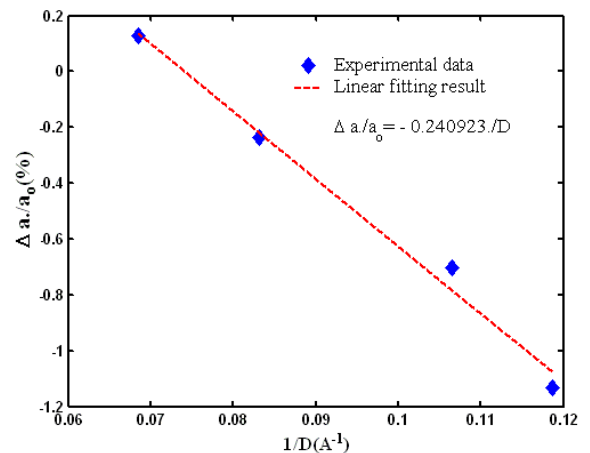


Fig. 3 variation of the relative lattice parameter Δa/a<sub>0</sub> as a function of the reciprocal diameter of Au nanoparticles embedded in alumina matrix

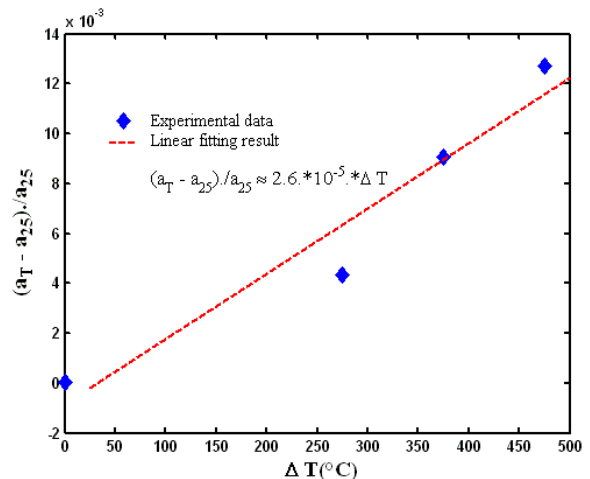


Fig. 4 The plots of gold lattice parameter, calculated from Au (111) diffraction versus temperature

An important aspect in this work is the thermal behavior that the metallic nanoparticle presents during heating. The lattice thermal expansion (α<sub>gold</sub>) whose value depends on the temperature can be calculated by the following relationship:

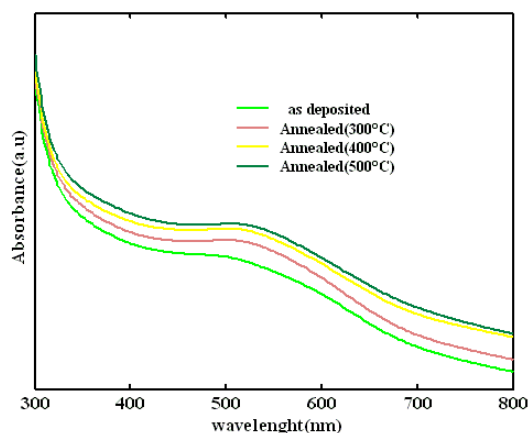
$$\frac{a_T - a_{25}}{a_{25}} = \alpha_{gold} \Delta T \tag{2}$$

Where T is the annealing temperature and ΔT = T - 25 °C.

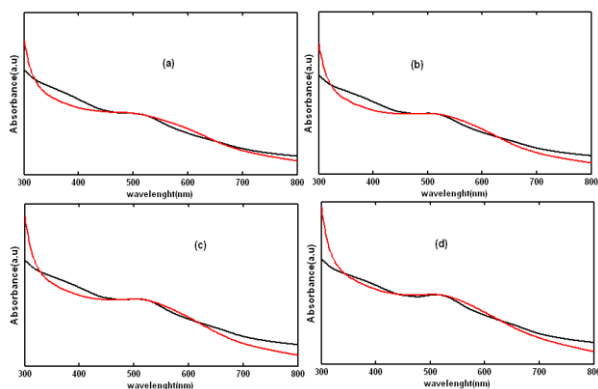
It can be seen from Fig. 4 that the evolution of lattice constant shows roughly linear relation with increasing temperature. The lattice thermal expansion coefficient obtained is 2.6 x 10<sup>-5</sup>/°C, which is a factor of 2 times larger than that of the bulk Au (1.4 x 10<sup>-5</sup>/°C).

### 3.2 Optical Characterization

The optical absorption spectra for the as-deposited and for the various heat-treated nanocomposite films are shown in Fig. 5. The spectra revealed an absorption in the short wavelength region of <450 nm, which is due to 5d to 6sp inter-band transition [22] and exhibit a broad absorption band at around 500 nm, which was assigned to the SPR of Au particles. The intensity of the SPR absorption band increased progressively with the increase of the annealing temperature of the films. This effect is attributed to the increment in the particle size caused as mentioned above by the increase of the annealing temperature of the samples, since the SPR absorption of small metal particles is originated from the collective motion of the conduction electrons interacted with external electromagnetic field of the incident radiation. The measured optical absorption curves were modeled based on Mie theory taking into account the mean free path of gold bulk conduction electrons. Using for gold nanoparticles the same values of plasma frequency, Fermi velocity, damping constant and the scattering parameter,  $w_p, v_F, \Gamma, A$ , respectively cited in the work [35–37]. Theoretical absorption spectra were also calculated by a combination of the same Eqs. (1) and (2) used in this work. The dielectric constant values of the bulk were taken from [42].



**Fig. 5** Optical absorption spectra of Au/Al<sub>2</sub>O<sub>3</sub> thin films as-deposited and at different annealing temperatures



**Fig. 6** Experimental and Mie simulated optical absorption spectra for the films as deposited (a) annealed at (b) 300 °C, (c) 400 °C and (d) 500 °C

Figs. 6(a–d) show the simulation and the experimental absorption spectra. The simulation results are summarized in Table 2. The plasmon peak positions just vary from 495 nm to 511 nm and the size increases slightly from 2.0 nm to 2.5 nm when heating temperature increases from 25 °C to 500 °C. The variation of SPR peak position ( $\lambda_{\max}$ ) with the annealing temperature is reported in Fig. 7. The solid line is a linear fit to the experimental data corresponding to the absorption maxima as determined from the calculated spectra in Fig. 6. An equation is derived in order to predict the maximum plasmon wavelength  $\lambda_{\max}$  at different temperatures.

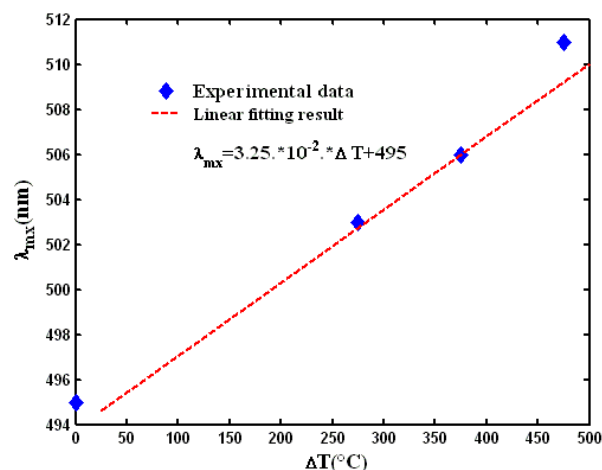
$$\lambda_{\max} = 3.25 \cdot 10^{-2} \Delta T + 495 \quad (3)$$

The value of 495 nm represents the SPR at room temperature. The Au nanoparticle sizes estimated from optical absorption spectra are larger than those measured from XRD. It is probably due to that the Scherrer's equation is based on size limited bulk structure and relating particle size to the peak width, cannot be used with great accuracy for very small clusters [43]. Formation of gold nanoparticles embedded in silica films

using RF-magnetron sputtering technique with subsequent thermal treatment in the same range of temperature have been reported in the work [36]. These authors found that after heat treatment the size and the lattice parameter increases, while the SPR red shift from 500 to 505 nm. In another work [35] the authors investigated the surface plasmon of gold clusters embedded in alumina matrix found that the peak position of the SPR exhibits a linear dependence with lattice parameter.

**Table 2** Plasmon peak position, size and dielectric functions of the surrounding medium of the sample as-deposited and heating at different temperatures

Annealing Temperature (°C)	SPR (nm)	Size (nm)	$\epsilon_m$
25	495	2	3
300	503	2.2	3
400	506	2.3	3
500	511	2.5	3



**Fig. 7** Variation of surface plasmon resonance peak position ( $\lambda_{\max}$ ) with annealing temperatures

The heat treatment annealing process is the most important and effective method to control the nanoparticle size. Independently of the production technique, several authors report different post-annealing effects. Hence, the temperature effects on the SPR absorption band of metal NPs were studied by Kreibitz [11, 44, 45] and Doremus [46, 47], and the origin of temperature effects upon the SPR was analyzed by Mulvaney [48]. Recently, the influence of temperature on SPR in Ag, Au and Cu NPs at high temperatures was reported in the works [49–52] and the results of a similar study of Au-based plasmonic nanostructures at low temperatures were reported in work [53], where appreciable red-shift and broadening of the SPR with increasing temperature were observed.

About the peak position of the SPR, except for the well-known blue shift effect of the SPR peak as a function of decreasing particles radius a redshift induced by the strong matrix effect has also been found by Smithard [54]. Kreibitz and Vollmer [11] pointed out the numerous experiments performed on Ag<sub>n</sub> clusters had proved the sensitivity of the size evolution of the Mie frequency on the surrounding matrix. Palpant and coworkers [55] investigated the surface plasmon energy of silver clusters in the framework of the time-dependent local density approximation (TDLDA), and found that the various observed size trends may be conclusively explained by simple dielectric effect. They also proved theoretically that the size effects are very sensitive to the matrix, the porosity at the interface.

### 4. Conclusion

In this work, thin films of Au-alumina nano composite deposited by co-sputtering technique have been investigated. XRD studies confirmed the presence of Au nanoparticles in amorphous alumina matrix. The lattice parameter exhibits a linear dependence with annealing temperature and thermal expansion coefficient of gold is deduced. Regular red shift in SPR with tunability from 495 to 511 nm is achieved after post heat treatment. The observed redshift of SPR varies linearly with annealing temperature is explained on the basis of increased particle size of Au nanoparticles and are in agreement with modified Mie theory.

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